

Table 2—Ferrite Measurements on 100% Ferrite Materials, Beam Balance Procedure

Sample number	Nominal composition	Calculated iron, %	TF with P5861 Magne-Gage no. 3 magnet, gms	EFN = 5.11(TF)
95928	29Cr-4Mo-2Ni	64.1	26.92	138
8-4638	29Cr-4Mo	66.4	28.63	146
1783C	25Cr-5Ni-3.5Mo	66.4	29.25	149
2125A	25Cr-3.5Mo	70.6	30.21	154
1274E	21Cr-3Mo	74.2	30.87	158
31606	18Cr-2Mo	77.7	32.39	166
U2	18Cr-2Mo	78.3	32.51	166
9317-448	13Cr-5Ti	86.2	34.09	174
Ingot iron	Fe	100.0	35.89	183

variable. Compositions of each sample were obtained from the supplier except for the 409 weld metal, which was run in the author's lab.

A very clear trend in TF (and EFN) was noted on these samples when the amount of iron in each sample was calculated by subtracting the known alloy composition from 100%. The TF and EFN decrease consistently with decreasing percent iron, although each sample is essentially 100% ferrite. The data are shown in Table 2 and Fig. 2. The relationship appears to be parabolic, although it can be approximated by a straight line over a limited range.

Samples of stainless steel casting alloys CF8 (308) and CF8M (316) were obtained from the Steel Founders' Society. These samples have varying compositions to cover a range of ferrite contents from 0 to 48 volume percent as determined metallographically by point counting. The samples and analytical methods are completely described in the SFS report (Ref. 6). It should be noted that point counting is considered a valid method of measuring true volume percent of ferrite in castings, because the ferrite is relatively coarse and tends to be at least partly spheroidized. Point counting has not proved suitable for weld metal, however, because the ferrite tends to be extremely fine and irregularly shaped.

The casting alloy samples had been polished on 2 or more faces for the prior point counting work, and it was not clear which face corresponded to which ferrite data point in the SFS report. So the procedure adopted was to make EFN measurements on each polished face of each sample, average the results for all the polished faces on a given sample, and compare the results with the averaged reported volume percent ferrite data for the given sample.

Direct FN Magne-Gage measurements using a no. 3 magnet were also made on the polished faces and averaged as well. In the process of making measurements of FN by the AWS procedure or EFN by the beam balance procedure, a slight tendency for the magnet to "hunt" (presumably seeking ferrite islands on the casting surface) was noted. The data

obtained as described above are given in Table 3.

It can be seen from Table 3 that the EFN values obtained by the beam balance procedure agree quite well with the FN values obtained with a Magne-Gage. It can also be seen that the FN and EFN values are numerically greater than the volume percent ferrite measured by point counting.

A graphical display of EFN vs. volume percent ferrite by point counting for the casting alloy samples shows a straight line relationship—Fig. 3. A regression fit of a straight line to the data of Fig. 3 shows a near-zero intercept, a slope of about 1.4 and a correlation of 0.99. This is an excellent fit and seems to indicate that the EFN (or FN) overestimates the volume percent ferrite in these alloys by about 40%.

Figure 2 indicates that the magnetic response of ferrite in iron-base material is strongly a function of the composition of that ferrite. In particular, the attractive force to a Magne-Gage no. 3 magnet is a function of the weight-percent of iron in the ferrite. One might know the composition of the ferrite, and there might be no effect of ferrite shape and distribution on its magnetic response. Then from the ratio of the magnetic response of an

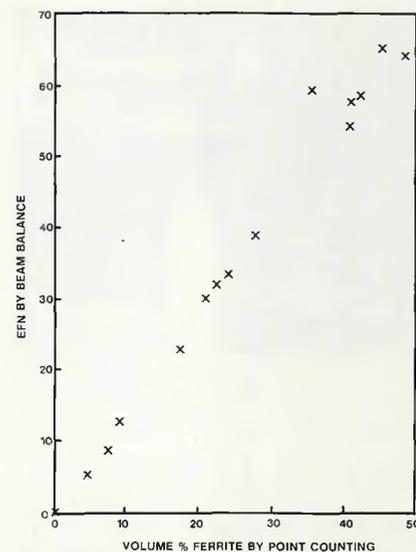


Fig. 3—EFN vs. volume-% ferrite for stainless casting alloys

unknown sample (its EFN, for example) to the magnetic response of a sample of 100% ferrite of the same composition as the ferrite in the unknown sample, one could directly calculate the volume percent ferrite in the unknown sample.

Solidification theory tells us that under ordinary conditions of cooling, duplex austenitic-ferritic materials experience a partitioning of alloying elements between ferrite and austenite. This occurs such that the composition of the ferrite in a duplex alloy is not the same as the overall alloy composition. For example, Lyman (Ref. 7) has shown by scanning transmission electron microscopy (STEM) that chromium and nickel in a Type 304L stainless weld metal of about 18.75 Cr, 9.75 Ni will redistribute themselves so that the ferrite is about 26 Cr, 4 Ni. However, it can be inferred from the data of Lyman that the iron content of the ferrite is nearly the same as that of the overall alloy despite the very pro-

Table 3—Ferrite Measurements on Stainless Casting Alloy Samples

Sample number	Average volume percent ferrite by point counting (Ref. 6)	Average FN by AWS procedure with Magne-Gage	Average beam balance TF, gms	Average EFN by beam balance procedure
45T	0.23	0.4	0.03	0.2
6882	4.40	5.8	0.99	5.1
6851	7.04	8.7	1.66	8.5
3492	8.98	12.2	2.44	12.5
3496	17.16	22.5	4.46	22.8
3491	20.85	Off scale	5.87	30.0
3485	22.46	Off scale	6.24	31.9
3494	23.94	Off scale	6.51	33.3
3499	27.98	Off scale	7.60	38.8
3631	35.56	Off scale	11.58	59.2
3488	40.78	Off scale	10.57	54.0
3611	41.30	Off scale	11.27	57.6
3501	42.43	Off scale	11.46	58.6
3489	45.45	Off scale	12.75	65.2
3495	48.64	Off scale	12.59	64.3

Table 4—Ferrite Ratios or NFN's For Cast Alloys

Sample no.	Alloy	Fe, %	Equation (3) EFN at 100% ferrite	EFN from Table 3	NFN = $\frac{100 \times \text{EFN}}{\text{EFN @ 100\% ferrite}}$	Avg. volume % ferrite by point counting
45T	CF8M	64.41	142.0	0.2	.14	.23
6882	CF8	68.34	149.0	5.1	3.42	4.40
6851	CF8M	66.27	145.3	8.5	5.85	7.04
3492	CF8M	67.12	146.8	12.5	8.51	8.98
3496	CF8	68.84	149.9	22.9	15.28	17.16
3491	CF8M	65.69	144.3	30.1	20.86	20.85
3485	CF8M	65.67	144.2	32.0	22.19	22.46
3494	CF8	69.38	150.8	33.4	22.15	23.94
3499	CF8	68.91	150.0	39.0	26.00	27.98
3631	CF8	70.04	152.0	59.4	39.08	35.56
3488	CF8M	65.70	144.3	54.2	37.56	40.78
3611	CF8M	64.16	141.6	57.8	40.82	41.30
3501	CF8	69.12	150.4	58.8	39.10	42.43
3489	CF8M	63.83	141.0	65.4	46.38	45.45
3495	CF8M	64.99	143.0	64.2	44.90	48.64

nounced redistribution of elements such as Cr and Ni.

If this is the case, then the information in Table 2 and Fig. 2 can be used to estimate the magnetic response or EFN of 100% ferrite having an iron content similar to the iron contents of the CF 8 and CF 8M castings. Then if the effect of ferrite shape and distribution on magnetic response were negligible, one could estimate the true volume fraction of ferrite in an unknown sample by the ratio of its EFN to the EFN of 100% ferrite of the same iron content.

Figure 2 gives the appearance of a slightly parabolic relationship between the EFN of 100% ferrite and the percent iron in the ferrite. The casting alloys of Table 3 cover a nominal iron content range from their compositions (given in Reference 6) of 63.83 to 70.04% iron. Over the range of about 64 to 78% iron, the data of Fig. 2 can be fit by a straight line with very good correlation. This regression line (omitting the ingot iron and 13 Cr-0.5 Ti points from the data of Table 2) is given by:

$$\text{EFN} = 28.1 + 1.77 (\% \text{ Fe}) \quad (3)$$

with a correlation of 0.9785. This line is shown on Fig. 2.

Using the composition data of Aubrey et al. (Ref. 6), equation (3) can be used to estimate the EFN of 100% ferrite of the same iron content as the casting alloys of Table 3. When this is done and the ratio of the measured EFN of each casting alloy to the EFN of 100% ferrite of the same iron content is calculated and converted to percent by multiplying by 100, the result is an almost perfect 1:1 correlation between that ratio converted to percent and the volume percent ferrite measured by point counting for each casting alloy. These results are given in Table 4 and shown graphically in Fig. 4.

The last result is rather exciting. This is because it seems to indicate that this ratio

converted to percent (which will hereafter be referred to as the Normalized FN or NFN) numerically agrees well with the volume percent ferrite of casting samples. However, the reader is cautioned that there is no assurance that such agreement would be obtained with weld metal since, as has been noted previously, point counting is unsuitable for measuring ferrite in weld metal due to the fineness and irregular shape of weld metal ferrite. Nevertheless, the results with the casting alloys encouraged the author to explore further the idea of Extended Ferrite Numbers and their possible appli-

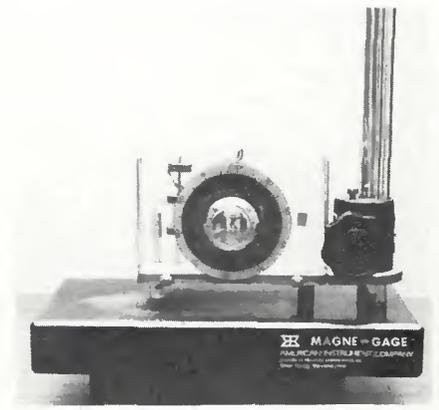


Fig. 5—A standard Magna-Gage. The white dial used for FN measurements is visible

cation to the measurement of stainless steel weld metal ferrite.

Method 2—EFN Measurements With a Magna-Gage

As noted previously, while a beam balance can be used to make FN measurements according to the IIV method or EFN measurements as described above, its use is inconvenient and subject to some interpretation of readings due to the tendency for very slight vibrations to cause premature detachment of the magnet from the specimen as the load is increased. Experience with the Magna-Gage over perhaps 10 years by numerous users has shown very little tendency

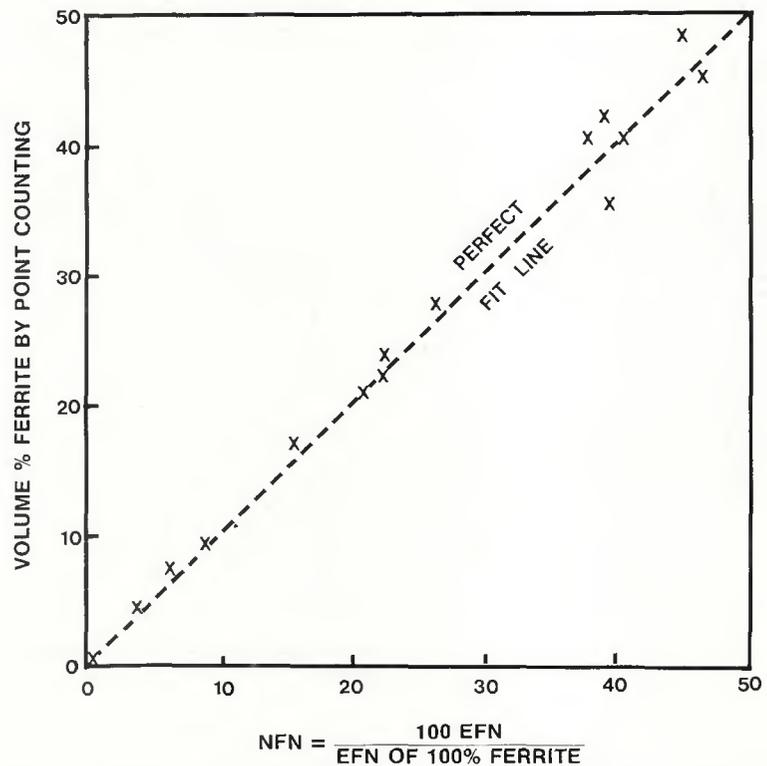


Fig. 4—Volume-% ferrite by point counting vs. normalized ferrite number (NFN) for CF8 and CF8M casting alloys

for premature detachment. This seems due to the design of the Magne-Gage which effectively damps out vibrations. Clearly, then, it would be desirable to use a Magne-Gage, if possible, for EFN measurements, in preference to use of a beam balance.

Discussions with the designers of the Magne-Gage indicated that the limitation of spring strength could be overcome by adding counterweights to the balance beam of the instrument. The spring and corresponding white dial readings will cover a range of about 5 to 6 grams of force. By adding appropriate counterweights, one can then cover any expected TF, provided one first determines a scaling factor for the counterweights and another for the white dial readings of the instrument. Figure 5 shows a Magne-Gage with its white dial scale.

A four-step procedure was devised to calibrate a Magne-Gage with its no. 3 magnet to obtain EFN values consistent with the beam balance method described previously. These steps are:

1. Develop a straight line relation between the Magne-Gage white dial reading (WD) and the force in grams (TF) required to pull the magnet from a specimen or raise the magnet with a known weight attached past the balance point of the Magne-Gage.

2. Develop a straight line relation pinned to zero between the TF and the Ferrite Number (FN) assigned to a set of eight NBS Coating Thickness Standards by AWS A4.2.

3. Determine the scale factor (M) between counterweights (W) hung from a small hole about 1½ in. (38 mm) from the Magne-Gage balance beam fulcrum

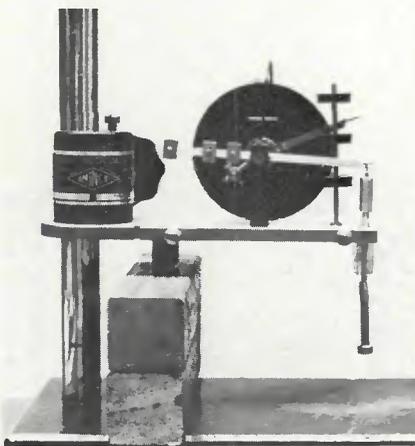


Fig. 6—Rear view of Magne-Gage with cover removed and a small weight hanging from the no. 3 magnet

Table 5—Weight or TF vs. WD for Magne-Gage P5861

Weight number	1	2	3	4	5	6	7	8
Weight, grams	0.00	0.50	1.02	1.98	2.78	3.90	4.56	5.46
WD	110	100	90	71	55	33	19	1

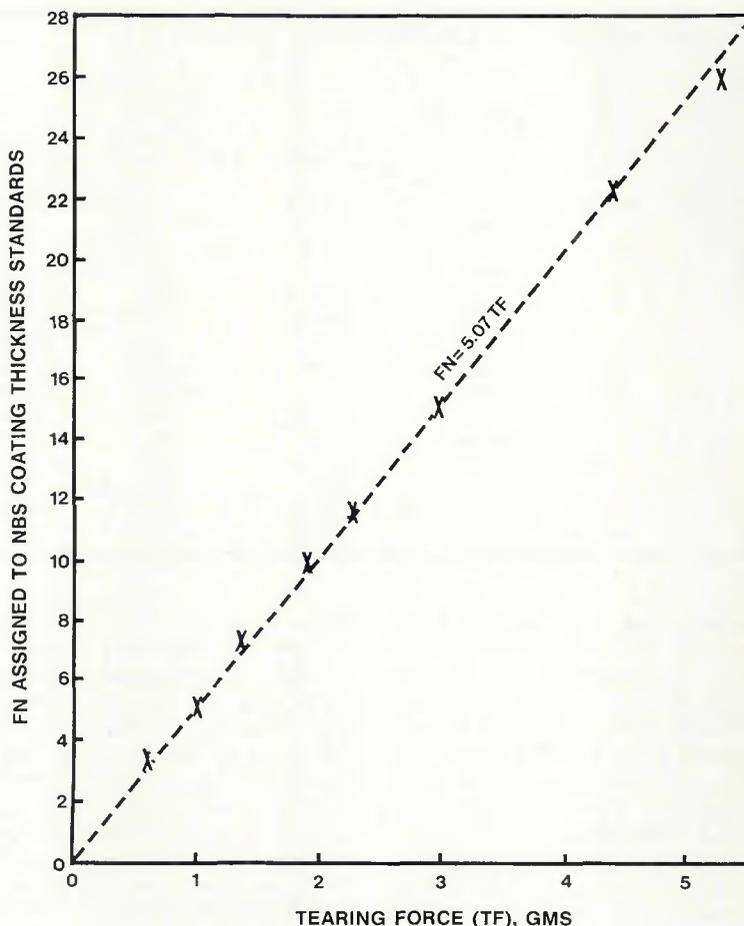


Fig. 7—FN vs. tearing force for Magne-Gage P5861

opposite the magnet hanger and weights hung from the magnet.

4. Combine the relations of steps 1, 2 and 3 to produce a relation of the EFN to the white dial reading (WD) and the added counterweight (W).

Once a calibration has been made, it is only necessary to check that the instrument calibration is unchanged according to AWS A4.2 before beginning to use it for EFN measurements, although a check on a fully ferritic sample of previously assigned EFN is recommended.

Details of the four-step calibration procedure follow.

Step 1. Prepare a set of iron-base magnetic weights covering the range of about ½ gram or less to about 6 grams in increments of 1 gram or less. Determine the weight of each to the nearest 0.01 gram. It is necessary that each weight be small enough at one end to fit into the hole in the plastic collar around the Magne-Gage magnet through which the magnet would contact a test specimen.

Hang each weight, in turn, from the Magne-Gage no. 3 magnet and adjust the white dial of the Magne-Gage so that the magnet is pulled out of the hole in the plastic collar a short distance. Then increase the spring tension by slowly turning the white dial clockwise until the spring tension exactly balances the weight. Read and record the white dial reading (WD) at this point along with the weight. This weight is equal to the force (TF) applied when the magnet is pulled from a weld specimen at this white dial reading.

Repeat this procedure for each of the weights. A table of white dial readings vs. weights will be obtained. Table 5 was so obtained with Magne-Gage no. P5861. Figure 6 shows the Magne-Gage with a small weight hung from the no. 3 magnet.

Then a straight line is fitted to this tabular data by the method of least squares. Many pocket calculators can do this simply. This relation will have the form:

$$TF = C_0 - C_1 (WD) \quad (4)$$

For the P5861 Magne-Gage, the regression line obtained with the above data is:

$$TF = 5.527 - 0.050 (WD) \quad (5)$$

tion (3) was used, for computing Normalized EFN values.

Reproducibility of the Magne-Gage Procedure

The utility of the WRC Ferrite Number system given in AWS A4.2 lies in the fact, established in countless tests, that Magne-Gage "A" at Company "X" will produce the same FN within a relatively small scatter band as Magne-Gage "B" at Company "Y". Table 9, which is reproduced from AWS A4.2-74, indicates that the instrument-to-instrument variation in Ferrite Number readings on a given weld metal sample will be on the order of $\pm 5\%$ of the average value for 95% of all instruments tested. It is this excellent reproducibility from instrument-to-instrument that makes the WRC Ferrite Number system the best available yardstick for measuring and/or specifying weld metal ferrite content.

It is, therefore, appropriate to consider the reproducibility of the EFN procedure developed herein. While the EFN procedure is based upon the WRC Ferrite Number system, it should be kept in mind that the EFN procedure is really an extrapolation of the FN system. Extrapolation tends to amplify errors. The author (Ref. 5) has shown that there is a very small systematic error from Magne-Gage to Magne-Gage that is linked to the strength of the magnet of the particular instrument.

A Magne-Gage with a stronger magnet tends to give a lower FN on weld metal than a Magne-Gage with a weaker magnet even though both Magne-Gages are calibrated to A4.2 using the same set of NBS coating thickness standards. As a

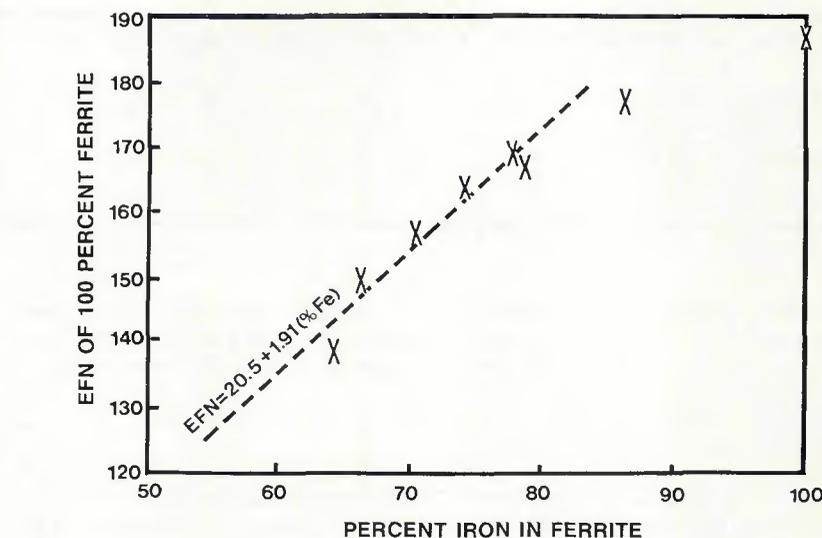


Fig. 9—EFN vs. percent iron in fully ferritic steel samples using Magne-Gage procedure, Magne-Gage P5861. Compare with Fig. 2

result, the IIW has found it appropriate to define a strength range for the magnet used in its procedure (Ref. 4). This strength range amounts to a calibration line slope as developed in equation (1) of 5 FN per gram (± 0.5). The AWS A4.2, on the other hand, sets no limit on magnet strength directly, relying upon the Magne-Gage manufacturer to supply magnets of appropriate strength. The magnets can lose strength with mishandling and possibly with age. This shifts the calibration line of the instrument sufficiently so that service by the manufacturer becomes necessary to meet the requirements of A4.2 and the magnet may be replaced at that time.

With these considerations in mind, the

author obtained from various sources five Magne-Gages in addition to the P5861 Magne-Gage used initially. The complete four-step calibration procedure described above was performed with each Magne-Gage. An EFN was obtained for each of the nine 100% ferrite steels of Table 8 with each Magne-Gage.

In addition to the nine 100% ferrite samples, a duplex weld metal sample was prepared from a specially-produced flux-cored filler metal. This alloy is one of the compositions given in the literature (Ref. 2) as suitable for stress-corrosion cracking resistance in chloride environments. The weld metal composition obtained with this flux-cored filler metal is given in Table 10. The alloy is, therefore, nominally 68.9% iron.

This duplex alloy weld metal served as a check on reproducibility of results from one instrument to another. The EFN values with each Magne-Gage on the nine 100% ferrite samples and on the duplex alloy weld metal are given in Table 11. In addition, an EFN for 100% ferrite of the same iron content as that of the duplex alloy weld metal was calculated for each Magne-Gage as done earlier with the cast alloys in the beam balance procedure. Then an NFN for the duplex alloy weld metal could be calculated as well. All of these results are also included in Table 11 along with the slope of the calibration line for each Magne-Gage.

It will be noted in Table 11 that the magnet strength of one Magne-Gage (P6443) is well outside the allowable strength range of the IIW procedure for FN measurements (though acceptable according to AWS A4.2). It should be noted that a large numerical value for the slope of the calibration line indicates a weak magnet, while a numerically small slope indicates a strong magnet. As predicted by the author's previous work (Ref. 5), the very weak magnet of P6443 produces numerically much larger values

Table 8—Ferrite Measurements on 100% Ferrite Samples with the Magne-Gage Procedure and, for Comparison, with the Beam Balance Procedure, in Both Cases using the P5861 No. 3 Magnet

Sample number	Nominal composition	Iron, %	EFN by Magne-Gage	EFN by beam balance (from Table 2)
95928	29Cr-4Mo-2Ni	64.1	139	138
8-4638	29Cr-4Mo	66.4	149	146
1783C	25Cr-5Ni-3.5Mo	66.4	149	149
2125A	25Cr-3.5Mo	70.6	156	154
1274E	21Cr-3Mo	74.2	163	158
31606	18Cr-2Mo	77.7	169	166
U2	18Cr-2Mo	78.3	168	166
9317-448	13Cr-.5Ti	86.2	177	174
Ingot iron	100 Fe	100.0	187	183

Table 9—Expected Range of Variation in Measurements with Calibrated Magne-Gages (reproduced from AWS A4.2-74)

Ferrite Number range	67% of the instruments	95% of the instruments
0-10	± 0.30 FN	± 0.60 FN
over 10 to 18	± 0.35 FN	± 0.70 FN
over 18	± 0.45 FN	± 0.90 FN

advised as to the suitability of using counterweights with the Magne-Gage. Access to Magne-Gages was provided by Babcock and Wilcox, Westinghouse and Alloy Rods Division of Allegheny Ludlum. To all of the above, the author is indebted.

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Discussion by H. C. Campbell*

Dr. Kotecki has postulated that each ferrite island in an austenitic stainless steel weld deposit or casting attracts a stan-

*H. C. CAMPBELL is a Consultant, Vero Beach, Florida, and a charter member of the AEC Advisory Committee on Type 347 Weld Metal, now the Welding Research Council Subcommittee on Welding Stainless Steel.

dardized Magne-Gage in proportion of its iron content. This is an attractive hypothesis.

His studies of fully ferritic weld metals show a quasi-linear relationship for half-a-dozen iron contents in the range that will be met in stainless weld metals. When he corrects his magnetic readings on ACl castings for their iron content, they agree nicely with the ferrite contents estimated by the Steel Founders' Society.

I like Dr. Kotecki's hypothesis. It needs refinement to pull in the outliers, but I predict it will survive critical examination.

The naturally occurring ferromagnetic elements are iron, cobalt, and nickel. I think the nickel content of a ferrite island may also contribute to its magnetic attraction. Dr. Kotecki's 100% ferrite specimens included only two nickel-bearing alloys; including a factor for nickel in the EFN formula for Extended Ferrite Number would much improve the fit of sample 1783C with 5 nickel; it would not improve the sample with 2 nickel. Ferritic samples with various iron and nickel contents should reveal a workable formula.

Ten years ago I encouraged Bill DeLong and my reluctant fellow members of the WRC Advisory Subcommittee for Stainless Steel to arbitrarily establish an innovative Ferrite Number System we had been slowly developing, making it the official measure of the Delta Ferrite Content of Austenitic Stainless Steel Weld Metal. I am delighted to see this encouraging agreement between our highly reproducible FN readings and point counts made by the Steel Founders' Society.

Author's Reply

Dr. Campbell's point regarding the effect of nickel on the magnetic attraction of ferrite is worth considering since many, but not all, nickel base alloys are ferromagnetic. The limited data of the present study is insufficient to reach a conclusion regarding this. However the work of Bungardt, Dietrich and Arntz cited in Reference 8 of the paper considers the effects of composition of ferrite on the field intensity for magnetic saturation of 100% ferrite stainless steels. They

arrive at the following relation between field intensity and composition:

$$4\pi I_F = 21,200 - 305 (\% \text{ Cr}) - 322 (\% \text{ Ni}) \pm 410 \text{ gauss}$$

This relation shows that both nickel and chromium have negative effects on the field intensity for magnetic saturation, and the magnitude of the effects of the two elements are similar. The difference in field intensity in the above relation for a 5% nickel change vs. a 5% chromium change would be far less than the error scatter band indicated. This supports my assumption that both nickel and chromium in place of iron in the ferrite reduce the magnetic attraction of the ferrite.

Subsequent to completion of the manuscript, I considered the EFN of 9% nickel steel. In the as-received condition, this material is martensite which is also ferromagnetic. It had a measured EFN of 178. Since I wanted to work with ferrite, I austenitized the 9 nickel steel at 1600°F (871°C), furnace cooled in 1 hour to 1300°F (704°C) (which is above the upper critical temperature) then furnace cooled at 10°F (5.6°C) per hour to 620°F (327°C) (the Ms is about 680°F (360°C) for this alloy) and air-cooled to room temperature. Then I cooled to -320°F (-196°C) in liquid nitrogen and reheated to room temperature to be certain that any retained austenite transformed to martensite.

The resulting microstructure is ferrite with perhaps 10-20% tempered martensite. The measured EFN on this material is 184 using the P5861 Magne-Gage. From Table 8 and Fig. 9 of the paper, since 9% nickel steel is approximately 90% iron, we could expect an EFN of 180 if iron content were the only factor affecting EFN. The agreement of experiment with prediction is not perfect but it is quite close.

Obviously, a more thorough investigation of the effects of various alloy elements on the magnetic response of ferrite could be undertaken. One might anticipate that silicon, since it is only half the atomic weight of iron, would have a larger effect while molybdenum, which is a much heavier atom, would have a smaller effect. However, such effects would not detract from the main utility of the Extended Ferrite Number system proposed, which is to provide a convenient yardstick for characterizing duplex austenite-ferrite stainless steels.

—CALL FOR PAPERS—

IIW CONFERENCE ON UNDERWATER WELDING

A Conference on Underwater Welding (including thermal cutting) will be held June 27-28, 1983, as part of the 36th Annual Assembly of the International Institute of Welding. The IIW Assembly will be conducted June 25 through July 2, 1983, in Trondheim, Norway.

The provisional program for the two-day Conference on Underwater Welding includes general survey sessions on June 27 covering: 1) Technology of Underwater Welding and 2) Metallurgy of Underwater Welding. Conference colloquia scheduled for June 28 include: 1) Wet and Dry Welding; 2) Repair and Other Applications; 3) Physical, Metallurgical and Mechanical Problems; and 4) Inspection and Performance. Conference proceedings will be published in book form and distributed to conference participants.

The following schedule has been set for authors interested in presenting a paper at the conference:

November 15, 1982: Author must submit proposal, including author's name, title of contribution, and a summary of approximately 150 words.

December 15, 1982: The Organizing Committee will report back to prospective speakers.

March 1, 1983: Manuscripts of accepted contributions must be received by the Norwegian Organizing Committee. Manuscripts should be typed and illustrations prepared according to instructions that will be mailed to authors by December 15, 1982. The paper should not exceed 10 pages in length.

Responses to this call for papers should be addressed to:

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